

Hydrogen induced room-temperature ferromagnetism in Co-doped ZnO: first-principles and Monte Carlo study

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Abstract The structural stability, vibrational and magnetic properties of hydrogen doped ZnO:Co have been studied by first-principles calculations based on density functional theory. Bond-center (BC) sites were identified to be most stable sites for hydrogen, the corresponding vibrational frequencies including anharmonic contributions were calculated. Its magnetic properties were investigated as well. The calculated results reveal that hydrogen could induce the change of electronic transfer, leading to a decrease of magnetic moment. However, the magnetic coupling between Co atoms is greatly strengthened. The results simulated by Monte Carlo method indicate that hydrogen can induce the Curie temperature to increase from 200 to 300 K.

Keywords Diluted magnetic semiconductors · First principles calculation · Monte Carlo simulation

1 Introduction

ZnO-based diluted magnetic semiconductors (DMS) have attracted considerable attention for their potential applications in spintronics and microelectronics in the few years [1]. Since Ueda's work on room temperature ferromagnetism (RTFM) for $\text{Zn}_{1-x}\text{Co}_x\text{O}$ was published [2], room-temperature ferromagnetism (FM) has commonly been

obtained in ZnO doped with Mn [3, 4], Co [5–8], Ni [9, 10], V [11–13], Cr [14–16] and Fe [17, 18]. However, the origin of ferromagnetism in Co-doped ZnO material is still controversial [19]. Some groups suggested that the observed FM was an intrinsic property [20, 21], while Risbud and Lawes et al. [22, 23] found no ferromagnetism in Co-doped polycrystalline ZnO. Moreover, the issue of hydrogen in ZnO was extensively investigated [24–28]. It is noteworthy that hydrogen is commonly present in the crystal growth environment and inevitably incorporates into ZnO crystal. First-principles calculations carried out by Van de Walle [29] revealed that interstitial hydrogen in ZnO acts as a shallow donor. Subsequently, theoretical and experimental efforts confirmed further the existence of the shallow donor hydrogen state in zinc oxide [30, 31]. Furthermore, it was suggested that H can mediate a strong short-range spin–spin interaction between magnetic ions in $\text{Zn}_{1-x}\text{Co}_x\text{O}$, leading a high temperature ferromagnetism [24]. Experimental studies indicated that H played an important role in the enhancement of ferromagnetic spin–spin interactions that went much beyond a carrier-mediated effect [25, 26]. Recently, the electrical and magnetic properties of low-energy H^+ -implanted ZnO single crystals with hydrogen concentrations up to ~3 at% have been investigated [31]. The hydrogen-induced ferromagnetism in H-ZnO samples is demonstrated by the magnetization and magnetotransport measurements [32]. Moreover, the magnetism of hydrogen-treated ZnO:Co was supported experimentally by magnetic measurements as well as theoretically by first-principles calculation [33]. They found that the two most favorable configurations are those where the H atoms reside at the Co-O bond center (BC) sites, namely BC_{\parallel} and BC_{\perp} . It was suggested that one possible mechanism for the ferromagnetism is hydrogen-facilitated interaction, and H plays an important role in inducing

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ferromagnetism. Therefore, it is crucial to get a deep understanding of the microscopic and electronic properties for hydrogen in Co-doped ZnO.

In this paper, we carry out first-principles calculation to study the structural stability of $\text{Zn}_{1-x}\text{Co}_x\text{O}$ and achieve the local vibrational modes of hydrogen-related complexes. It is significant to identify the microscopic configuration and helpful to experimental studies by infrared and Raman spectroscopy. Moreover, the magnetic properties of $\text{Zn}_{1-x}\text{Co}_x\text{O}$ induced by hydrogen are investigated. Finally, Monte Carlo method was employed to calculate Curie temperatures of hydrogen-doped ZnO-based DMSs.

2 Model and calculation

2.1 First-principles calculations

The total energy and vibrational frequencies calculations have been performed using density functional theory (DFT) [34] with the projected augmented wave (PAW) [35] potentials as implemented in the Vienna *ab initio* simulation package (VASP) [36, 37]. The cutoff energy for the plane wave expansion of electron wavefunction was set at 520 eV. A gamma-centered $3 \times 3 \times 2$ k mesh was adopted to sample the irreducible Brillouin zone for $3 \times 3 \times 2$ ZnO supercell. All atoms in each doped supercell were fully relaxed using the conjugate-gradient algorithm until the maximum force on a single atom was less than 0.02 eV/Å.

Schematic representation for possible hydrogen sites in wurtzite ZnO is shown in Fig. 1. As in ZnO, interstitial H acts as a shallow donor (H^+) [38]. AB_N (antibonding Nitrogen) and BC (bond center) configurations can be distinguished for the wurtzite structures [39]. There exist two types of configurations for AB_N and BC: one type is long to the c axis (labeled as BC_\parallel and $\text{AB}_{\text{N},\parallel}$), the other

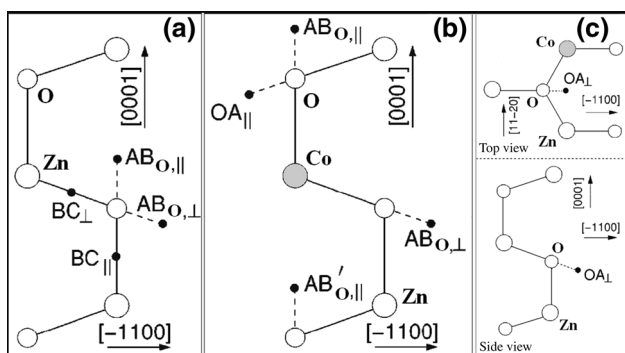


Fig. 1 Schematic representation of possible hydrogen sites in the (11–20) plane of wurtzite ZnO

type is “perpendicular” to the c axis (labeled as BC_{\perp} and $\text{AB}_{\text{N},\perp}$). Considering the possibility of H near to dopant, OA_{\parallel} , OA_{\perp} and $\text{AB}'_{\text{N},\parallel}$ sites are proposed as well [40].

The oscillation potential of H, $V(x)$, can be described as follows:

$$V(x) = \frac{k}{2}x^2 + \alpha x^3 + \beta x^4, \quad (1)$$

where the coefficient of the quadratic term gives the harmonic frequency $\omega^0 = \sqrt{k/\mu}$, where reduced mass μ is defined as: $\frac{1}{\mu} = \frac{1}{m_{\text{H}}} + \frac{1}{m_{\text{O}}}$, here m_{H} and m_{O} are the masses of the hydrogen and oxygen atoms, respectively.

Higher order coefficients α and β describe anharmonic contributions. Moreover, the Schrödinger equation of one-dimensional single-particle is described by

$$\left[-\frac{\hbar^2}{2\mu} \nabla^2 + V(x) \right] \psi(x) = E\psi(x). \quad (2)$$

Based on the perturbation theory [40, 41], an approximate analytical solution for the equation above can be obtained as follows:

$$\omega = \omega^0 + \Delta\omega = \sqrt{\frac{k}{\mu}} - 3\frac{\hbar}{\mu} \left[\frac{5}{2} \left(\frac{\alpha}{k} \right)^2 - \frac{\beta}{k} \right], \quad (3)$$

where ω is the total frequency, ω^0 is the harmonic frequency, and $\Delta\omega$ is the anharmonic contribution.

2.2 Monte Carlo method

The Curie temperatures for magnetic study will be calculated using the Monte Carlo simulation with the magnetic atoms distributed randomly. We construct $L \times L \times L$ ($L=16, 20$ and 32) wurtzite ZnO with periodic boundary conditions. The ratios of ferromagnetic Co and non-magnetic Zn atoms are assumed to be x and $(1-x)$, respectively. Meanwhile, it is assumed that the Co atoms are distributed randomly on the Zn lattice sites of wurtzite ZnO.

The Heisenberg Hamiltonian of the system will be described as

$$E = - \sum_{ij} J_{ij} \bar{S}_i \cdot \bar{S}_j - K \sum_i (\bar{S}_i \cdot \bar{u}_i) - H \sum_i S_i^z, \quad (4)$$

where J_{ij} is the exchange coupling constant. In this study, we label one Co atom as ‘0’ with its neighbor Co atom labeled from 1 to 7, as can be seen in Fig. 2. $J_{0j} = J_{ij}$ ($j = 1, 2, 3, 4, 5, 6, 7$) represents the exchange coupling constant between the i th and j th Co atoms. The first, second and third terms in Eq. (4) are the exchange interaction energy, anisotropic energy, and Zeeman energy, respectively [42, 43]. The thermodynamic magnetization per atom and the susceptibility can be calculated by

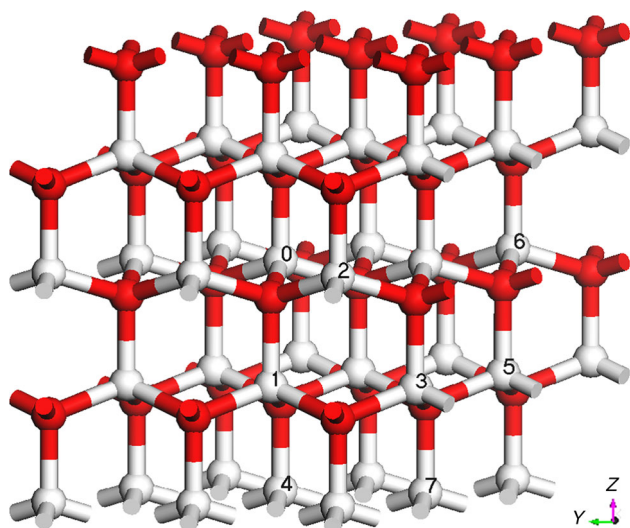


Fig. 2 (Color online) The $3 \times 3 \times 2$ supercell of wurtzite ZnO containing 36 Zn atoms (light) and 36 O atoms (dark), and 7 Zn atoms in different sites are labeled

$$M(T) = \left\langle \left[\left(\sum_i S_i^x \right)^2 + \left(\sum_i S_i^y \right)^2 + \left(\sum_i S_i^z \right)^2 \right]^{1/2} \right\rangle / N, \quad (5)$$

$$\chi(T) = N(\langle M^2 \rangle - \langle M \rangle^2) / T, \quad (6)$$

where N is the number of the magnetic Co atoms. In order to define the Curie temperature, an accumulation of magnetization of the fourth order U_L is described as follows

$$U_L(T) = 1 - \langle M^4 \rangle / 3 \langle M^2 \rangle^2. \quad (7)$$

The maximum slope in U_L from the T dependence can be used for evaluation of the transition temperature. The χ - T and U_L - T curves are used to define the Curie temperature [44, 45]. In the simulation, we set $K = 0$, $H = 0.00086$ eV.

3 Results and discussion

We first examine the energetics of H incorporation in Co-doped ZnO. One Zn atom substituted by Co in $3 \times 3 \times 2$ ZnO was investigated to identify the stable configuration. The calculated total energy differences ΔE , bond lengths of O–H and vibrational frequencies for different configurations of O–H in Co doped ZnO are listed in Table 1. From the table, it is found that H favors the bond-center (BC) sites, and BC_{\perp} is the most stable site while BC_{\parallel} is the metastable state with $\Delta E = 0.02$ eV. This result is consistent with the previous prediction [24, 29, 32]. The relaxed bond length of O–H for all the candidate configurations are approximate 1 Å.

Table 1 The calculated total energy differences ΔE , bond lengths of O–H and vibrational frequencies for different configurations of O–H, and the lowest energy is set as zero reference value

Sites	ΔE (eV)	d_{O-H} (Å)	Mag (μ_B)	ω^0 (cm^{-1})	$\Delta\omega$ (cm^{-1})	ω (cm^{-1})
BC_{\perp}	0.00	0.997	3.0000	3774	−271	3503
BC_{\parallel}	0.20	1.006	3.0001	3720	−258	3462
OA_{\perp}	0.34	0.990	3.0167			
OA_{\parallel}	0.35	0.991	3.0234			
$AB'_{O,\parallel}$	0.41	0.992	3.0227			
$AB_{O,\perp}$	0.42	0.990	3.0002			
$AB_{O,\parallel}$	0.45	0.990	3.0015			

Considering the anharmonic contribution, we get the vibrational frequencies with 3503 and 3462 cm^{-1} for H at BC_{\perp} and BC_{\parallel} site, respectively. Clearly, the calculated results of structural stability for H-doped $Zn_{1-x}Co_xO$ are significant to experimentally probe, being crucial to our subsequent study of magnetic property.

We first investigate the Co-doped ZnO without hydrogen doping, getting a magnetic moment of $3.07 \mu_B$. Then, from Table 1 it is found that the magnetic moments of all configurations slightly reduced when hydrogen is embedded, especially for BC_{\perp} site. It is obvious that hydrogen will induce a change of magnetism in Co-doped ZnO. Thus, a deep study of electronic transfer and magnetic coupling will be carried out.

A $3 \times 3 \times 2$ supercell of ZnO with one Zn atom substituted by Co atom is shown in Fig. 3, corresponding to a doped concentration of 2.78 %. A hydrogen atom is doped

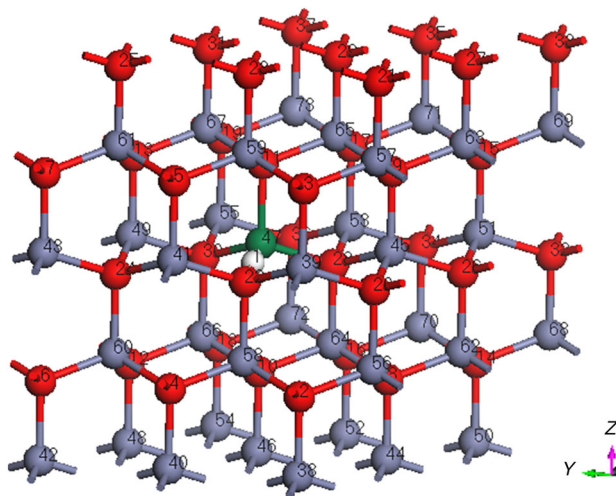


Fig. 3 (Color online) $3 \times 3 \times 2$ supercell of ZnO with hydrogen in the interstitial BC_{\perp} site. All the atoms are labeled from 1 to 73, containing 36 Zn atoms (light), 36 O atoms (dark), 1 Co atom (numbered 47) and 1 H atom (numbered 1)

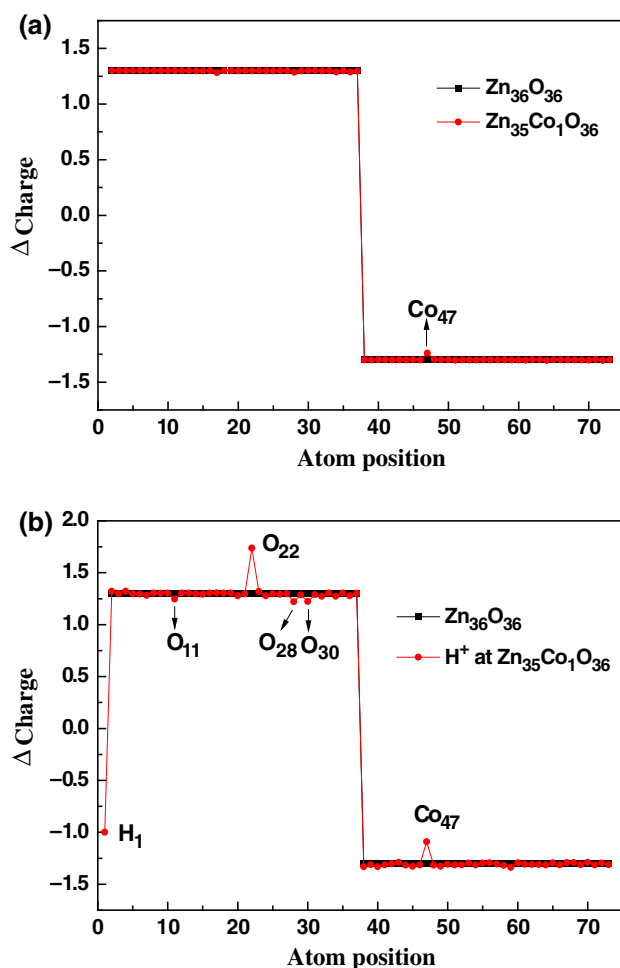


Fig. 4 (Color online) The electronic transfer structures for **a** $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 2.78\%$) and pure ZnO; **b** $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 2.78\%$) with H^+ in BC_\perp site and pure ZnO

in the most stable interstitial site BC_\perp . All the 73 atoms are labeled from 1 to 73. Figure 4(a) and (b) show the electron transfer structures of single Co doped ZnO without and with hydrogen doped, respectively. From Fig. 4(a), it is found that for pure ZnO each Zn atom loses about 1.30 electrons while each O atom gets about 1.30 electrons. For the $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 2.78\%$), Co atom has a obvious electron transfer and loses 1.24 electrons, which results in a slight electron transfer of its neighbor O atoms. From Fig. 4(b), it is observed that as hydrogen is doped, the electronic structure has a significant change. Due to the electrical activity, hydrogen readily incorporates with the 22nd O atom. The hydrogen loses 1.0 electrons, inducing a increase of electronic transfer for O₂₂ atom to 1.74. Moreover, compared to Fig. 4(a), it is found that the number of the lost electrons for the other three O atoms (numbered 11, 28 and 30) bonding with Co is decreased. Co atom loses only 1.09 electrons, which gives rise to the weaken of the coupling between Co and O₂₂ atoms.

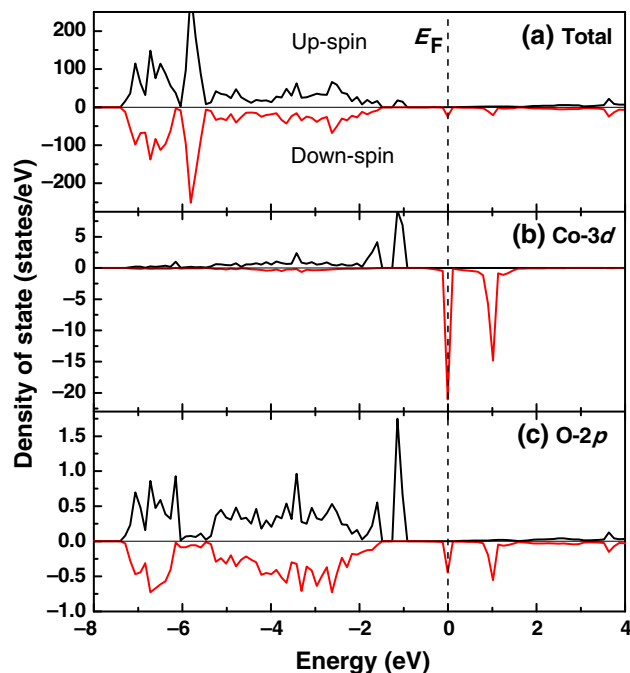


Fig. 5 (Color online) **a** Total density of states (DOS) and **b** Co-3d and **c** O-2p projected density of states (PDOS) of $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 2.78\%$) ($3 \times 3 \times 2$ supercell) without hydrogen doped. The Fermi level (dashed line) is set as the zero energy. Positive (negative) values correspond to the majority (minority) spin

The total density of states (DOS) and projected density of states (PDOS) of $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 2.78\%$) ($3 \times 3 \times 2$ supercell) without hydrogen doping are shown in Fig. 5, respectively. From the figure, a half-metallic behavior can be found with the majority spin being semiconducting and minority spin being metallic. The spins of conduction electrons at the Fermi level are almost 100% polarized. Comparing the total DOS with the PDOS of the Co atom, it is found that most of the spin polarization states originate from Co-3d electrons. In addition, a hybridization between the Co-3d and O-2p bands is observed in the vicinity of the Fermi level. Therefore, the strong coupling between O-2p and Co-3d is mainly responsible for the ferromagnetic ground state in Co-doped ZnO before hydrogen is doped. Figure 6 shows the total density (DOS) and projected density of states (PDOS) of $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 2.78\%$) when one hydrogen atom is doped. Unlike the hybridization by Co-3d and O-2p in Fig. 5, the spin polarization states of DOS at Fermi level originate from the coupling of H-1s and Co-3d bands. It is observed that the presence of hydrogen can increase the carriers concentration of system, which is similar to the results obtained by Roberts et al. [46]. Hydrogen atom tends to transfer electrons to the minority spin state of Co-3d, leading to a decrease of magnetic moment on Co ion.

To study the magnetic coupling, two Co atoms substituting for Zn are considered in $3 \times 3 \times 2$ ZnO, corresponding

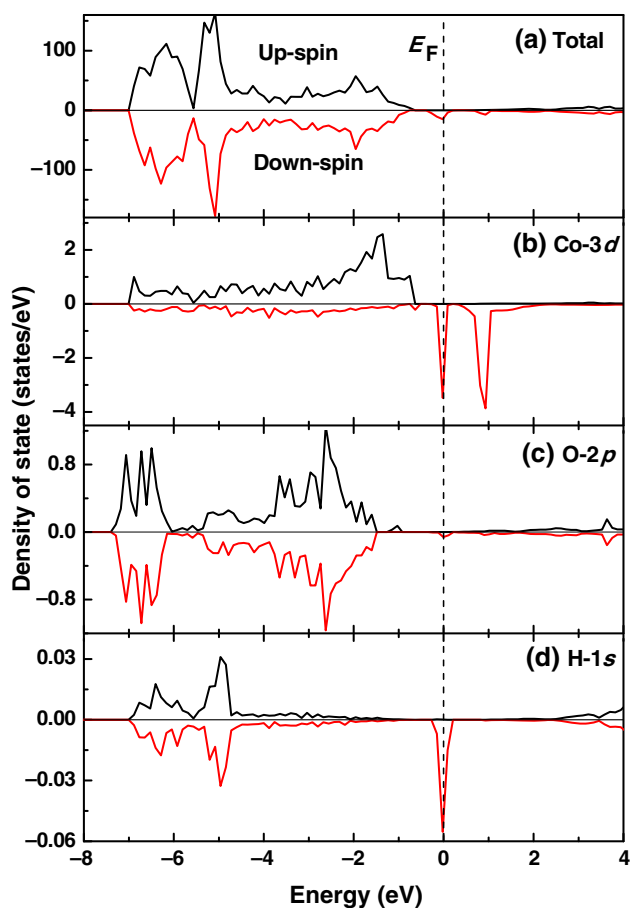


Fig. 6 (Color online) **a** Total density of states (DOS) and **b** Co-3d and **c** O-2p and **d** H-1s projected density of states (PDOS) of $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 2.78\%$) ($3 \times 3 \times 2$ supercell) with hydrogen doped. The Fermi level (dashed line) is set as the zero energy. Positive (negative) values correspond to the majority (minority) spin

to a doped concentration of 5.55%. We explore seven different configurations for Co atoms substitute at Zn sites marked (0, 1), (0, 2), (0, 3), (0, 4), (0, 5), (0, 6) and (0, 7), as seen in Fig. 2, respectively. Hydrogen atom is then present at interstitial BC_\perp site near to the Co atom labeled 0. For each configuration, the total energies of the ferromagnetic (E_{FM}) and antiferromagnetic (E_{AFM}) spin configurations are calculated. The magnetic coupling strength J for the pair of Co atoms is then obtained from the energy difference between the FM and AFM configurations ($J = E_{\text{AFM}} - E_{\text{FM}}$). The magnetic coupling strengths will be also used as input parameters for Monte Carlo simulation. Figure 7 shows the magnetic coupling strength of the $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 5.55\%$) for different configurations before and after hydrogen is doped. From the figure, it is found that the system has an FM ground state for all the configurations, which is consistent with previous reports [43, 47]. When hydrogen is doped, although the magnetic moment of system slightly decrease from $6.06 \mu_{\text{B}}$ to $5.59 \mu_{\text{B}}$, the

values of magnetic coupling strength J are found to be noticeably increased. Moreover, as the concentration of H^+ is enhanced, the magnetic moment of system greatly declines. The change trend is shown in Fig. 8, where the nearest neighbor configuration (0, 1) and the next neighbor configuration (0, 2) are considered due to their stronger Co-Co interaction among seven configurations. Figure 9(a) and (b) show the calculated magnetization M and U_L by Monte Carlo simulation for $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 5.55\%$) without and with hydrogen doped, respectively. From the figures, it is observed that the Curie temperature T_c of $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 5.55\%$) is about 200 K, near to experimental results [2, 48]. As hydrogen is

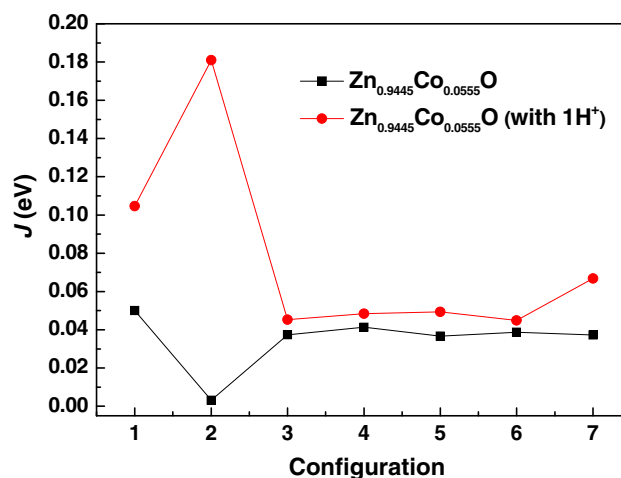


Fig. 7 (Color online) Magnetic coupling strength J between two substitutional Co of 7 configurations before and after hydrogen is doped

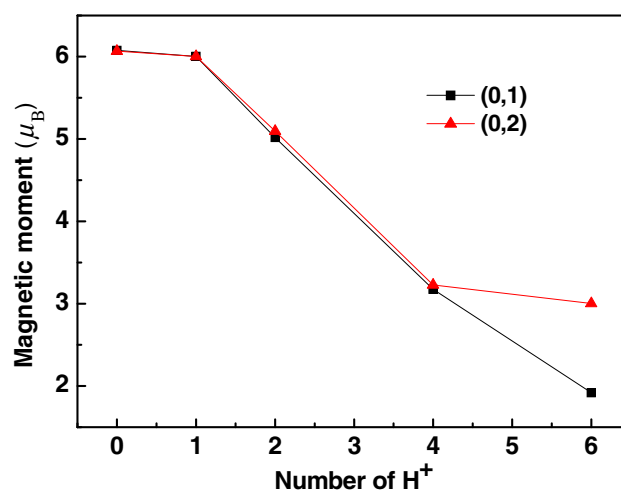


Fig. 8 (Color online) Magnetic moment as a function of number of doped H^+ for nearest neighbor configuration (0,1) and next nearest neighbor configuration (0,2) in the $\text{Zn}_{1-x}\text{Co}_x\text{O}$ ($x = 5.55\%$) system

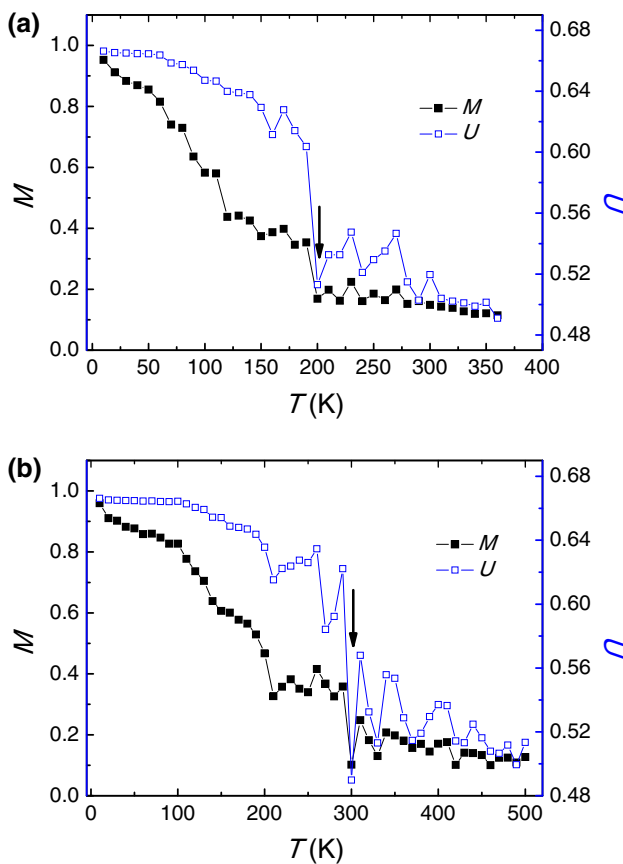


Fig. 9 (Color online) The simulated magnetization M and U_L as a function of temperature. **a** Without hydrogen doped, the T_c is 200 K. **b** With hydrogen doped, the T_c is 300 K

doped, a strong H-mediated spin-spin interaction can be induced [24], which leads to high temperature ferromagnetism with $T_c > 300$ K.

4 Conclusions

In this paper, the first-principle calculation and Monte Carlo simulation were employed to study the Co-doped ZnO diluted magnetic semiconductor with hydrogen doped. Bond-center (BC) sites were identified to be most stable sites for hydrogen, and the corresponding vibrational frequencies were calculated. The magnetic properties were investigated as well. The results reveal that hydrogen could lead to a change in electronic transfer, inducing the magnetic coupling changes, resulting in the increase of the Curie temperature from 200 to 300 K.

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